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# Density effect of gold nanodisks on the SERS intensity for a highly sensitive detection of chemical molecules

Jean-François Bryche · Raymond Gillibert · Grégory Barbillon · Mitradeep Sarkar · Anne-Lise Coutrot · Frédéric Hamouda · Abdelhanin Aassime · Julien Moreau · Marc Lamy de la Chapelle · Bernard Bartenlian · Michael Canya

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Abstract Surface Enhanced Raman Scattering is a sensitive and widely used as spectroscopic technique for chemical and biological structure analysis. One of the keys to increase the sensitivity of SERS sensors is to use nanoparticles/nanostructures. Here, we report on the density effect of gold nanodisks on SERS intensity for a highly sensitive detection of chemical molecules. Various densities of gold nanodisks with a height of 30 nm on gold/glass substrate were fabricated by electron beam lithography in order to have a good uniformity and reproducibility. The evolution of the Enhancement Factor with nanodisk density was quantified and compared to numerical calculations. An enhancement factor as high as  $2.6 \times 10^7$  was measured for the nanodisk with a diameter of 110 nm and a periodicity of 150 nm which corresponds to the biggest density (42.2%).

**Keywords** Surface Plasmons  $\cdot$  E-beam Lithography  $\cdot$  Biosensors  $\cdot$  SERS  $\cdot$  Plasmonics

### 1 Introduction

Surface Enhanced Raman spectroscopy (SERS) is a powerful technique of molecular detection. The SERS effect

Jean-François Bryche · Grégory Barbillon · Frédéric Hamouda · Abdelhanin Aassime · Bernard Bartenlian Institut d'Electronique Fondamentale CNRS UMR 8622, Université Paris-Sud, Bâtiment 220, 91405 Orsay Cedex, France. E-mail: jean-francois.bryche@u-psud.fr

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Raymond Gillibert · Marc Lamy de la Chapelle Laboratoire Chimie, Structures, Propriétés de Biomatériaux et d'Agents Thérapeutiques, CNRS UMR 7244, Université Paris-Nord, 93017 Bobigny Cedex, France. is strongly dependent on the electromagnetic (EM) field intensity. In order to compensate for the very low Raman scattering cross-sections of the target molecules, EM enhancement is necessary and can be obtained with plasmonic nanostructures. As SERS signals strongly depend on the size, the shape and the organization of the nanostructures, the uniformity, reproducibility and stability of SERS substrates with high enhancement factor are extremely important [1,2]. These are critical bottlenecks in the realization of industrial SERS applications. Several lithographic techniques, such as Nanoimprint Lithography (NIL) [3–5], NanoSphere Lithography (NSL) [6–8], Electron Beam Lithography (EBL)[9– 13, X-ray Interference Lithography [14] are used to fabricate SERS substrates. Recently, we also demonstrated the presence of a hybrid plasmonic mode when a gold film under the nanocylinders is added [15]. This plasmonic mode allows a strong enhancement of the electromagnetic field intensity, and consequently a potential improvement of the SERS signal. Following this study, we report in this paper on the density effect of gold nanodisks on the SERS intensity. Experimental results are shown for nanodisk arrays of different periods and compared to numerical calculations. To ensure a good uniformity and reproducibility, electron beam lithography was used to fabricate the samples.

#### 2 Experimental details

#### 2.1 Au nanodisk fabrication

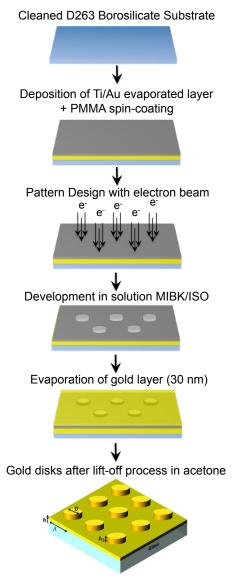
The fabrication process is summarized in figure 1. Main steps are (i) cleaning of the glass substrate, (ii) evaporation of titanium/gold layers under vacuum and PMMA spin-coating, (iii) E-Beam Lithography, (iv) develop-

ment (v) second gold deposition by evaporation under vacuum, (vi) lift-off process.

Substrates are 900  $\mu m$  thick D263 borosilicate glasses from Schott with a refractive index of 1.53 at  $\lambda=546$  nm and a transmission around 90% in IR-Visible domains. The borosilicate glass is first carefully cleaned following a classical protocol, which includes solvents and  $H_2SO_4/H_2O_2$  (3:1) solution (Piranha solution). After drying, the sample is covered with 2 nm Ti / 30 nm Au layers obtained by electron beam evaporation. Titanium is used here as adhesion layer.

To prepare the e-beam lithography step, PMMA A2 (polymethylmethacrylate) is spin-coated and baked on the substrate (80 nm thick). Several  $100 \times 100 \ \mu m^2$  and  $200 \times 200 \ \mu m^2$  gold nanodisk arrays with periodicities  $\Lambda$ ranging from 80 to 300 nm and diameters D of 50 and  $110 \ nm$  are fabricated. E-Beam parameters have been carefully optimized to take in account the over-exposure resulting from the proximity effects. As the pattern density chosen is rather important, obtaining a good definition of the nanodisks (i.e. shape, dimensions and periodicity), leads to set the writing current between 2 and  $2.2 \, nA$  with a dose between 6 and  $7.5 \, nC/cm^2$ . All samples were realized with a NanoBeam (nB4) sys-tem with an accelerating voltage of 80 kV. Samples are developed in a (1:3) methylisobutylketone/isopropanol (MIBK/IPA) solution during about 2 min, and finally rinsed with isopropanol. A 30 nm gold layer is then evaporated on the sample, and the nanodisks are fi-nally obtained by lift-off process in acetone. Eventually, to be sure that resist residues are completely avoided, the sample is totally cleaned by oxygen plasma during 5 min at 200 W power.

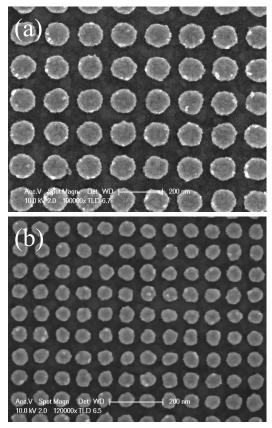
On figure 2, we show SEM pictures for two couples of parameters ( $D = 110 \ nm$ ,  $\Lambda = 150 \ nm$  and D = 50nm,  $\Lambda = 80 nm$ ). The nanodisks obtained on the figure 2 show good definition with an average diameter of 113 nm and 52 nm compared to the defined values, respectively. In the same way, the average gap of 37 nm, and 29 nm are obtained for these two types of patterns, respectively. No significant dispersion of these parameters was observed across the whole pattern area. We obtain for these couples of parameters, two fill-factors (or surface densities) defined by the ratio between the disk area (top) and the square of the periodicity:  $\rho =$  $(\pi D^2)/(4\Lambda^2)$ . Thus, we have around 42.2% for the first couple (110-150 nm) and 30.7% for the second (50-80 nm) compared to the maximum density of 78.5% (for disks in contact).



**Fig. 1** Principle scheme of the fabrication of gold nanostructures used for chemical sensors,  $h_1$  and  $h_2$  represent the gold film thickness  $(h_1 = 30 \ nm)$ , and the nanostructure height  $(h_2 = 30 \ nm)$ , respectively.

### 2.2 Thiophenol functionalization

For the sample functionalization, thiophenol  $(C_6H_6S)$  molecule is used in order to test the sensitivity of these samples as SERS substrates. The functionalization is decomposed in four steps: (i) preparation of a 0.1 mM solution of thiophenol in ethanol, (ii) dipping the sample in the solution during 2 h, (iii) washing thoroughly with pure ethanol, (iv) drying with nitrogen. In addition, the thiophenol concentration used for the Raman experiment is 1M.



**Fig. 2** SEM pictures of gold nanodisks on gold surface (scale bar = 200 nm) with the following sizes : (a) D=110 nm,  $\Lambda=150$  nm, and (b) D=50 nm,  $\Lambda=80$  nm.

# 2.3 Optical characterization

Raman spectra were recorded using a Xplora spectrophotometer from Horiba Scientific. The acquisition time was fixed to 10 s. A 660 nm laser was used for all experiments with a power of 288  $\mu W$ . The laser excitation was focused on the substrate using a microscope objective ( $\times 100$ , N.A. = 0.9). The same objective was used to collect the Raman signal from the SERS substrates in a backscattering configuration. SERS spectra were recorded with a spectral resolution under  $4 \text{ cm}^{-1}$ . The illuminated area is around 144  $\mu m^2$ . In addition, for classical Raman measurements in solution, a macroobjective with a focal length of 40 mm (N.A. = 0.18) and a 660 nm laser were used. All obtained spectra have been normalized by the acquisition time. We also take into account the N.A. and the laser power in order to determine the E.F.

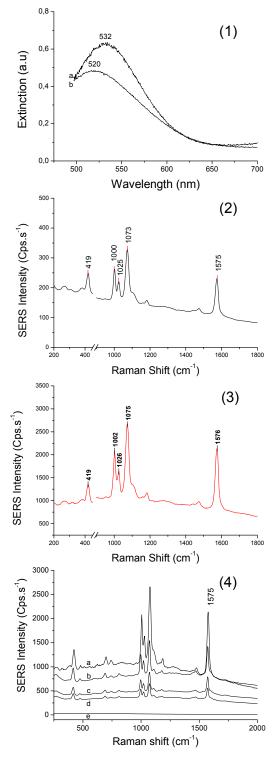
#### 3 Results and discussion

Functionalized nanodisks with the thiophenol molecules following the protocol described in the paragraph 2.2

were characterized by SERS measurements. Figure 3(1) represents the extinction spectra of nanodisks for two different couples of dimensions: (a)  $D=50~nm,~\Lambda=80~nm,~$  and (b)  $D=110~nm,~\Lambda=200~nm.$  The plasmonic resonances associated to these two couples of dimensions are  $\lambda_{Plasmon}=520~nm,~\lambda_{Plasmon}=532~nm,~$  respectively. Figures 3(2) and 3(3) represent the SERS spectra of thiophenol molecules obtained with these two samples. Characteristic Raman shifts of thiophenol molecules [16] are observed in the figures 3(2) and 3(3). No significant SERS signal is observed for the gold film (see figure 3(4e)). The highest values of SERS intensities were found for the size D=110~nm and  $\Lambda=150~nm$  (see figure 3(3)).

On figure 3(4), we studied the effect of the grating's periodicity on the SERS intensity by keeping constant the diameter of disks (D = 110 nm). A reduction of the SERS intensity is observed for all characteristic peaks, when the periodicity of nanodisks increases (see Table 1, for the values of the peak intensity at 1575  $cm^{-1}$ ). This is explained by the reduction of the coupling between nanodisks and the consequent reduction of the electric field intensity around them. Proximity between nanodisks plays a key role for SERS signal, as well as the nanodisk size. Similar studies based on nanodisks fabricated on Si substrates have been reported before. Such reports have demonstrated SERS enhancement either by structures fabricated by X-ray interference lithography [14] with small gap or by simulation [18]. However in such structures with a metallic array on glass or Si substrates, only the localized plasmon resonance (LSPR) can be excited. Thus, the reported enhancement results from the confinement of the electromagnetic field by such LSPR modes. The structure and the physical phenomenon studied in this article is however different from the above mentioned citation, in the fact that the structure studied here consists of an underlying uniform metallic layer along with the metallic array. This uniform layer can support the propagating surface plasmon (PSP) resonance. The PSP owing to the periodicity of the structure can couple to give rise to hybrid modes [15,17]. Such hybrid modes of-fer a greater enhancement of the electromagnetic field around the metallic cylinders. Thus, the SERS inten-sity obtained by using an array of metallic cylinders with an underlying uniform metallic film is higher than using the array on Si [14,19] or only on glass [20] sub-strates. Moreover, to compare efficiency of our substrates, we evaluate the Enhancement Factor (EF), which is defined as follows:

$$EF = \frac{I_{SERS}}{I_{Raman}} \times \frac{N_{Raman}}{N_{SERS}} \tag{1}$$



**Fig. 3** (1) Extinction spectra of nanodisks for: (a) D=50  $nm,~\Lambda=80$   $nm,~\lambda_{Plasmon}=520$  nm, and (b) D=110  $nm,~\Lambda=200$   $nm,~\lambda_{Plasmon}=532$  nm. SERS results for nanodisks with the following sizes: (2) D=50 nm and  $\Lambda=80$  nm, (3) D=110 nm and  $\Lambda=150$  nm. Characteristic Raman shifts are at 419 / 1000 / 1025 / 1075 and 1575  $cm^{-1}$ . (4) SERS spectra for nanodisks with a diameter of 110 nm and four periods: (a) 150 nm, (b) 200 nm, (c) 250 nm, and (d) 300 nm, (e) no SERS signal is observed from the smooth gold film.

where  $I_{SERS}$ ,  $I_{Raman}$  are the SERS and Raman intensities, respectively.  $N_{SERS}$ ,  $N_{Raman}$  are the number of excited molecules in SERS and Raman experiments, respectively.  $N_{SERS}$  depends on the numerical aperture and the laser spot size. It also depends on the periodicity and the size of the nanoparticles.  $N_{SERS}$  is calculated from the following equation:

$$N_{SERS} = N_A \times S_{Collected} \times \sigma_{Surf} \times \frac{S_{Structures}}{\Lambda^2}$$
 (2)

where  $N_A$  is the Avogadro's number  $(mol^{-1})$ , then  $S_{Collected} \approx 144 \ \mu m^2$  (see Section 2.3),  $S_{Structures}$  is the sum of the lateral and top surfaces of the nanocylinder.  $\sigma_{Surf}$  is the surface coverage of thiophenol, which is approximately  $0.544 \ nmol/cm^2$  [21]. In the Raman experiment, the number of excited molecules  $N_{Raman}$  is  $4.22 \times 10^{15}$ . This number is obtained by [22]:  $N_{Raman}$  $= N_A \times C \times V$ , where C is the used concentration of Thiophenol (1M), and V is the scattering volume and equal to 7 nL. A maximum experimental EF value of  $2.6 \times 10^7$  is found (see Table 1) for our system with the highest density (diameter of 110 nm, and periodicity of 150 nm). When the periodicity increases from 200 nm to 300 nm, the EF value decreases from 2  $\times$  $10^7$  to  $1.5 \times 10^7$ . Also, the standard deviation (DS) on a same substrate by taking into account the dose used for the lithography, is given in the table 1. The sampleto-sample variation has the same order of magnitude. We can directly link EF decreasing with the reduction of the nanodisk density.

ſ	D(nm)	$\Lambda$ $(nm)$	ρ (%)	EF	DS (%)
ſ	110	150	42.2	$2.6 \times 10^{7}$	11.7
İ	110	200	23.8	$2 \times 10^{7}$	9.1
Ì	110	250	15.2	$1.6 \times 10^{7}$	13.5
Ì	110	300	10.6	$1.5 \times 10^{7}$	15.4
	50	80	30.7	$1.9 \times 10^{6}$	14.2

**Table 1** For the peak at  $1575 cm^{-1}$ , Enhancement Factor (EF) for different fill-factors  $\rho$ .

In addition, for small size of nanodisks (diameter of 50 nm and periodicity of 80 nm), an EF of 1.9 ×  $10^6$  is found. We suppose that this weaker value can be explained by the position of the experimental surface plasmon resonance ( $\lambda_{Plasmon} = 520 nm$ , see figure 3(1a)) that is far from the excitation wavelength ( $\lambda_{exc} = 660 nm$ ) with such small nanoparticles, and also by the smaller density of this structure compared to the sample with 110 nm nanodisks (30.7 vs 42.2%).

As shown elsewhere in the literature [23,24], the SERS intensity follows the structure local field enhancement at the excitation frequency and the Stokes shifted

emission frequencies. This is the origin of the common  $|E^4|$  approximation [25] for SERS enhancement factor. To estimate the SERS EF we therefore need to evaluate the near field intensity at the excitation and emission frequencies, respectively. Rigorous numerical calculations were carried out for this purpose to verify the trends observed in the experimental SERS EF. The numerical model [15,26] that hybridizes two popular numerical methods namely the finite element method (FEM) and the Fourier modal method (FMM), was used for the calculation of near field intensities in the structures. Using these results, the SERS intensity distribution over the surface of the structure can be estimated as follows:

$$I_{SERS} = \beta \times I_{\lambda_0}(x, y, z) \times I_{\lambda_R}(x, y, z)$$
 (3)

where  $\beta$  is a space independent constant and depends on the Raman tensor and thus on the chemical properties of the molecules used.  $I(\lambda_0)$  and  $I(\lambda_R)$  are the near field intensities  $(E_x^2+E_y^2+E_z^2)$  at the excitation wavelength  $\lambda_0$  and  $\lambda_R$ , respectively. In figure 4, we show the SERS intensity distribution of the structures for periods of 150 nm and 300 nm, considering plane wave incidence normal to the structures. The SERS intensity was normalized to the incident intensity  $(I_0)$ for both the calculated intensities at the incident and emission wavelengths. We see that the SERS intensity decreases as the period of the structures increases as observed by the experimental study. To find the SERS Enhancement Factor of the structure, the previous local SERS intenisty must be integrated around the whole cylinder. The calculated SERS enhancement factor can

$$EF_{EM} \approx \frac{1}{S} \times \iint_{S} I_{SERS} \, dS$$
 (4)

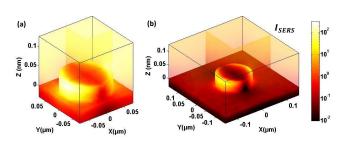


Fig. 4 The calculated SERS intensity distribution in log scale for two periods (a)  $150\ nm$  and (b)  $300\ nm$ 

The integral is normalized by the surface area over which the integration was carried out. The SERS intensity for two periods with the 110 nm diameter is displayed on figure 4. To compare these results, we define simulated ratios  $R_{Field} = EF_{EM_{110-X}}/EF_{EM_{110-300}}$  and experimental ratio  $R_{EF} = EF_{110-X}/EF_{110-300}$  reported in Table 2 with the periodicity of 300 nm as reference and X the studied period from 150 nm to 300 nm.

D(nm)	$\Lambda$ $(nm)$	$R_{Field}$	$R_{EF}$
110	150	1.7	1.7
110	200	1.1	1.3
110	250	1.03	1.06
110	300	1	1

Table 2 For the peak at  $1575 \ cm^{-1}$ , comparison of the trend of calculated SERS electromagnetic field intensity and experimental SERS EF.

A good agreement between simulation and experiment is obtained on density effect of nanodisks.

#### 4 Conclusion

In this paper, we demonstrated the density effect of gold nanodisks on the SERS intensity with different densities varying from 10.6% to 42.2% on a gold film. As expected, the highest density of nanodisks (42.2%) gave the greatest value of EF (EF =  $2.6 \times 10^7$ ) due to a better coupling between nanodisks and the consequent increasing of the electric field intensity around them. These experimental results are confirmed by numerical calculations of the electromagnetic field intensity. In addition, the size, and shape of nanostructures are homogenous and reproducible on large pattern areas, which are important points for industrial applications.

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