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Systematic behavior of electro-optic chromophore photostability

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The wavelength dependence of the one-photon absorption-induced photodegradation rate has been measured from the visible to the near IR for a variety of electro-optic chromophore-doped polymers. Systematic behavior is identified. The lifetime of the electro-optic activity is found to increase exponentially over 4–6 orders of magnitude for wavelengths ranging from peak of absorption, typically in the visible, to \( \sim 1000 \) nm. Many popular chromophores developed for electro-optics over the past 10 years are compared.

During the past decade certain aspects of polymers that contain chromophores have been investigated intensively, and the performance of polymers required for a variety of photonics applications has improved steadily.\(^6\) However, to use these highly efficient materials one must address their stability. The temporal stability of the chromophore orientation necessary for electro-optic activity to be sustained has been investigated intensively.\(^3\) Photostability has been less well explored, and the reported studies have often been made over limited wavelength ranges, usually only in the visible or the infrared.\(^4-9\)

The same chromophores being studied for electro-optics are also used near their absorption maxima for holography, data storage, writing waveguides and gratings, etc. based on low-power, optically induced transformations to a new species that is effectively not electro-optically active. The temporal evolution under irradiation can easily be modeled.\(^11\) This schematic two-step process results in a cumulative loss of the material’s nonlinearity with time as progressively more of the active chromophores are removed. Our recent studies of 4-dimethylamino-4′-nitrostilbene (DANS) and 4-N-ethyl-N-(2-hydroxyethyl)amino-4′-nitroazobenzene (DR1) have shown that the dominant degradation mechanism is photo-oxidation.\(^12,13\)

The local rate of creation of the electro-optic inactive photoproduct \( N_2 \) is expressed as \( \partial N_2/\partial t = B^{-1}(\lambda) \partial N_E/\partial t = [\sigma(\nu)/B(\nu)] N_1(t) n(t) \), where \( N_E \) is the number of electro-optic chromophores in their excited state and \( N_1(t), n(t) \), and \( \sigma(\nu) \) are, respectively, the local electro-optic active chromophore concentration, the photon flux, and the molecular absorbptivity. Thus, the key parameter that describes the decay time of the population of the active species is \( \tau = B/\sigma n \), where \( B/\sigma \) is the photostability figure of merit. The larger \( B/\sigma \), the more stable the doped polymer material.

We investigated more than 40 chromophores, and representative examples of their chemical structures are listed in Table 1. Except for the side-chain polymers (SCP’s) DANS and DR1, all chromophores were studied as guest–host material systems. Each
material was deposited as a film onto glass substrates \( \sim 2 \, \mu m \) thick by use of spin coating. Photodegradation was induced by illumination with strong pump beams from 430 to 1300 nm. For charge-transfer chromophores the macroscopic electro-optic activity is proportional to the quantity of active chromophores present and hence to the strength of the charge-transfer absorption feature. We determined the concentration of chromophores by monitoring with a 543.5-nm probe beam (He–Ne laser, weak enough not to induce photodegradation) the temporal evolution of the transmission change. The probe beam’s transmission initially increases with time (i.e., the integrated flux), and the initial slope of the transmission change directly yields the ratio \( B/\sigma \). We verified that the evolution of the degradation curves depended solely on the integrated flux at different average pump power levels.

Representative results for \( (B/\sigma)(\nu) \) indicative of the trends measured in all the materials are shown in Fig. 1. Four clear trends emerge: (1) an increase in \( B/\sigma \) (and hence in chromophore lifetime) over orders of magnitude with decreasing photon energy \( h\nu \); (2) a distinct linear dependence of \( \log(B/\sigma) \) on \( h\nu \) for a few tens of percent above the absorption maximum \( h\nu_{\text{max}} \) (associated with the ground state to charge-transfer state transition) to 1.25 eV; (3) a relative decrease in \( B/\sigma \) near 1 eV (1320 nm), possibly owing to singlet oxygen as identified previously; and (4) a relatively constant \( B/\sigma \) for \( h\nu > h\nu_{\text{max}} \).

We have precise absorption measurements over the full wavelength range only for the stilbene SCP DANS (weight fraction, 43%). The inverse absorption coefficient \( \sigma^{-1} \) \( (\sigma \propto \alpha) \) exhibits (Fig. 2) an approximately logarithmic behavior relative to photon energy over the range 550–1000 nm, the same spectral region in which \( \log(B/\sigma) \) is linear in photon energy. There \( \sigma(\nu) \) dominates the wavelength dependence of \( B/\sigma \). The same \( \log(B/\sigma) \) dependence was found in the other chromophores studied. The plateau in \( B/\sigma \) near \( h\nu_{\text{max}} \) is a consequence of \( \sigma(\nu) \) varying by only a factor of 2 over the full half-width (100–200 nm) of the absorption spectrum. Note that \( B/\sigma \) typically rises with further increase in \( \nu \) on the high-energy side of \( r_{\text{max}} \), as expected; e.g., chromophore 4 in Table 1. The situation in DANS for \( \nu > r_{\text{max}} \) is more complex because of the presence of a second, higher-energy charge-transfer state with a different \( B \).

Figure 3 shows the dependence of \( B \) on \( h\nu \) for a few cases. Typically \( B \) is independent of wavelength. However, in the SCP DANS there are two distinct values of \( B \) that we associate with photodegradation of two distinct charge-transfer states, located at 4.17 and 2.90 eV. \( B \) varies by 4 orders of magnitude from chromophore to chromophore. The larger \( B \), the more stable the chromophore.

The log-linear behavior observed for \( B/\sigma \) implies that

\[
B/\sigma = D_0 \exp(-h\nu/E_0),
\]

where \( D_0 \) and \( E_0 \) are constants. This characteristic behavior allows \( B/\sigma \) to be estimated over the relevant wavelength range from measurements of \( B/\sigma \) at just two different values of \( \nu \). In a plot of \( D_0 \) versus \( E_0 \) (Fig. 4), there is a small spread in \( E_0 \) clustered about 0.085 eV, in contrast to \( D_0 \), which varies over 7 orders of magnitude, which implies that even one measurement at one \( \nu \) permits a good estimate over the whole visible–near-IR spectral range. As discussed above, \( E_0 \) depends on the absorption spectrum on the low-energy side. Our overall results indicate only weak dependence was found in the other chromophores.

### Table 1. Molecular Structures Studied at CREOL of Chromophores as Guests in Poly(Methyl Methacrylate) Host B Values in the Plateau Region, Location of \( \lambda_{\text{max}} \) and Symbols Used for Each in Figs. 1–4

<table>
<thead>
<tr>
<th>Number</th>
<th>Structure</th>
<th>( \lambda_{\text{max}} )</th>
<th>Symbols</th>
</tr>
</thead>
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<tr>
<td>1</td>
<td><img src="image1.png" alt="Structure 1" /></td>
<td>430</td>
<td>□</td>
</tr>
<tr>
<td>2</td>
<td><img src="image2.png" alt="Structure 2" /></td>
<td>475</td>
<td>○</td>
</tr>
<tr>
<td>3</td>
<td><img src="image3.png" alt="Structure 3" /></td>
<td>470</td>
<td>△</td>
</tr>
<tr>
<td>4</td>
<td><img src="image4.png" alt="Structure 4" /></td>
<td>640</td>
<td>▼</td>
</tr>
<tr>
<td>5</td>
<td><img src="image5.png" alt="Structure 5" /></td>
<td>510</td>
<td>◊</td>
</tr>
<tr>
<td>6</td>
<td><img src="image6.png" alt="Structure 6" /></td>
<td>520</td>
<td>+</td>
</tr>
<tr>
<td>7</td>
<td><img src="image7.png" alt="Structure 7" /></td>
<td>524</td>
<td>*4 × 10^6</td>
</tr>
<tr>
<td>8</td>
<td><img src="image8.png" alt="Structure 8" /></td>
<td>529</td>
<td>*5 × 10^7</td>
</tr>
</tbody>
</table>

*An asterisk identifies a value taken at Gemfire at 5 nm only.*

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Fig. 1. \( \log(B/\sigma) \) versus photon energy, \( E_{\text{photon}} = h\nu_{\text{pump}} \), for a representative range of materials listed in Table 1.
Fig. 2. $\alpha^{-1}$ of DANS, measured as film transmission in the visible and as waveguide propagation loss in the IR. The experimental data are fitted by use of a Voigt profile. Absorption for $\lambda > 1000$ nm shows maxima owing to vibrational overtones.

Fig. 3. Photostability parameter $B$ as a function of excitation photon energy, $E_{\text{photon}} = h\nu_{\text{pump}}$, for the SCP’s DANS (■) and DR1 (●) and the guest–host cases indicated by the various types of points.

Fig. 4. Values of $D_0$ and $E_0$ for the polymers listed for Fig. 3.

dependence of $E_0$ on the details of the chromophore structure, the host matrix, and whether the polymer is side chain or guest–host. In contrast, the prefactor $D_0$ depends strongly on the details of the molecular structure, the host, etc.

A prime application of such polymers is to electro-optic channel waveguide devices expected to operate for 10 years under continuous illumination.$^5$ Assuming a waveguide area of $10 \mu m^2$ and a 1-mW power level, $B/\sigma \approx 10^{15}$ is needed to reach this lifetime. This requires $B \approx 10^9$, already reached for solid-state dye laser molecules.$^5$ Some of the current materials studied come within 1 order of magnitude of this value.

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References